1   Revision 1

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3   Protoenstatite: A new mineral in Oregon sunstones with "watermelon" colors

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Abstract

Al-Fe-bearing protoentatite was discovered in Oregon sunstones with unusual pleochroic / dichroic red to green coloration using high-resolution transmission electron microscopy (HRTEM) and X-ray energy dispersive spectroscopy (EDS). The empirical formula calculated on the basis of 6 O *apfu* is $(Mg_{1.17}Fe_{0.43}Al_{0.26}Ca_{0.03}Na_{0.10}Ti_{0.01})_{\Sigma 2.00}(Si_{1.83}Al_{0.17})_{\Sigma 2.00}O_6$. The protoenstatite has a space group of *Pbcn*; its unit-cell parameters refined from selected-area electron diffraction patterns are $a = 9.25(1)$ Å, $b = 8.78(1)$ Å, and $c = 5.32(1)$ Å. The esds on the cell parameters were determined based on electron diffraction patterns from the coexisting native copper inclusion and the host labradorite with known cell parameters. Protoenstatite nanocrystals are quenchable to low temperature. The crystallographically-oriented nanocrystals of protoenstatite and clinoenstatite in association with copper nanocrystals are responsible for the unusual green and "watermelon" coloration of the labradorite gemstone.

Keywords: Oregon sunstone, labradorite, new pyroxene, clinoenstatite, protoenstatite, HRTEM, native copper, dichroic

Introduction

Enstatite, $Mg_2Si_2O_6$, with a space group of *Pbca*, has several polymorphic counterparts including clinoenstatite (*P2₁/c*), high-temperature clinoenstatite (*C2/c*), high-pressure clinoenstatite (*C2/c*), protoenstatite (*Pbcn*), and high-pressure protoenstatite (*P2₁cn*) (Cameron and Papike 1981; Tribaudino et al., 2002; Angel et al., 1992; Yang et al., 1999). Protenstatite is reported to be a high temperature form that cannot be quenched to room temperature (Cameron and Papike 1981; Tribaudino et al., 2002). Protoenstatite would transform to enstatite or clinoenstatite at low temperature based on the results of synthetic protoenstatite (Cameron and Papike, 1981; Chen and Prensnall, 1975; Smith, 1969; Smyth, 1974). However, a synthetic Li-Sc-bearing protoenstatite with smaller cations of Li and Sc in octahedral sites is quenchable at low temperature (Smyth and Ito, 1975). It is reported that protoenstatite was a precursor of clinoenstatite in some Mg-rich basalts (Dallwitz et al., 1966; Shiraki et al., 1980) and even in star dusts (Schmitz and Brenker, 2008). In this letter, results from electron diffraction and high-resolution TEM (HRTEM) imaging are presented. The mineral and name have been approved by Commission on New Minerals, Nomenclature and Classification (CNMNC) of the International Mineralogical Association (IMA 2016-117). Two characterized specimens (catalogue numbers UWGM 3538 and UWGM 3539) are deposited in the collections of the Geology Museum, Department of Geoscience, University of Wisconsin-Madison (1215 West Dayton Street, Madison, WI 53706, USA).

Samples and Experimental Methods

Protoenstatite occurs as precipitates associated with copper nanocrystals in gem-quality labradorite phenocrysts (Oregon sunstones) from Dust Devil Mine, Lake County, Oregon

62   (Figures 1 and 2). The site is located in the Rabbit Basin within the Oregon high desert. The host

63   rock is a mid-Miocene basalt (Johnson et al., 1991; Peterson, 1972). The phenocrysts are

64   generally tabular plates and large laths ranging from ~1 cm in the greatest dimension to lathes

65   8.3 cm long, 2.6 cm wide and 1 cm thick (Hofmeister and Rossman, 1985; Johnson et al., 1991;

66   Peterson, 1972; Stewart et al. 1966). Protoenstatite was discovered in the green part of the

67   "watermelon" variety that possesses a clear rim and a core of transparent red surrounded by a

68   clear vibrant green border that can only be seen in certain orientations (Figure 1). Some carved

69   or faceted red, green and watermelon Oregon sunstones are illustrated in supplementary material

70   (Fig. S1).

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72   The "watermelon" sunstones exhibit pleochroism and dichroism. Color and schiller are

73   always localized in the cores of the phenocrysts where the native copper micro- or nano-platelets

74   that populate the interior are exsolved clusters of crystals. They are most commonly exsolved

75   parallel to the feldspar crystal plane (010), but also (001) (Hofmeister and Rossman, 1985). This

76   dichroism is exhibited in hand samples of "watermelon" sunstones. The crystals exhibit

77   dichroism with a clear red when oriented approximately parallel to the feldspar (001) plane, and

78   both red and green are seen when oriented in [100] and [010] directions (Figures 1). The parts

79   with green color also exhibit a brownish red / green pleochroism under a plane polarized light

80   (Figure 2). The same phenomenon was observed by Johnson et al. (1991). Pleochroism appears

81   to become stronger in deeply colored specimens. By contrast, the clear rim does not show

82   pleochroism.

83   Because protoenstatite occurs as nano-inclusions in gem-quality "watermelon" sunstones,

84   ion-milled TEM specimens (on Mo grids) were used for the mineralogical characterization using

85   transmission electron microscope associated with an X-ray energy-dispersive spectroscopy (EDS)

86   system. HRTEM imaging, X-ray EDS and selected-area electron diffraction (SAED) analyses

87   were carried out using a Philips CM200-UT microscope equipped with GE light element energy-

88   dispersive X-ray EDS at the Materials Science Center, University of Wisconsin-Madison, and

89   operated at 200 kV. Chemical analyses were obtained using the EDS (spot size 5 with a beam

90   diameter of ~50 nm). Quantitative EDS results were obtained using experimentally determined

91   k-factors from standards of albite, forsterite, anorthite, orthoclase, labradorite, fayalite, and

92   titanite. The same method was used for characterizing nanocrystals of luogufengite, Al-bearing ε-

93   Fe_2O_3, (Xu et al., 2017).

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96   **Results and Discussion**

97   TEM images show the protoenstatite nano-precipitates within labradorite together with

98   native copper nanocrystals (Fig. 3). SAED patterns reveal their proto-pyroxene structure with the

99   space group *Pbcn* (Fig. 4). X-ray EDS analysis confirms their pyroxene stoichiometry (Fig. 5,

100   Table 1). The empirical formula calculated on the basis of 6 O *apfu* is

101   $(Mg_{1.17}Fe_{0.43}Al_{0.26}Ca_{0.03}Na_{0.10}Ti_{0.01})_{\sum 2.00}(Si_{1.83}Al_{0.17})_{\sum 2.00}O_6$. HRTEM image shows (100) lattice

102   fringes with a periodicity of 9.25 Å corresponding to periodic changes of skews of octahedral

103   layers along the *a*-axis (Fig. 6). The observed HRTEM image matches protoenstatite structure,

104   instead of enstatite or clinoenstatite (See Table 2 for comparison).

105   Unit cell parameters were determined based on diffraction patterns from the

106   protoenstatite nanocrystals. Neighbouring copper nanocrystals and the labradorite host were used

107   as internal standards. Unit cell parameters of the labradorite with a very similar composition

108   (from Lake County, Oregon) are from Wenk et al. (1980). The measured unit-cell parameters are

109    $a = 9.25(1)$ Å, $b = 8.78(1)$ Å, and $c = 5.32(1)$ Å. Calculated density of protoenstatite is 3.30

110   g·cm^{-3}. Calculated powder X-ray diffraction peaks are listed in Supplementary Material (Table

111   S1). Comparison among the enstatite polymorphs with pyroxene structures is listed in table 2.

112   The crystals larger than ~ 200 nm transformed into clinoenstatite with a high density of

113   stacking faults (Fig. S5), which is very similar to the observed microstructures in a fast-cooled

114   protoenstatite (Ijima and Buseck, 1975). Cooling of the lava resulted in transformations from

115   protoenstatite to clinoenstatite with a high density of stacking faults in large protoenstatite

116   crystals (> 200 nm), whereas small protoenstatite crystals (< 200nm) are preserved in the host

117   labradorite phenocrysts. The Al-bearing protoenstatite nanocrystals with large surface areas may

118   lower the phase transformation temperature and stabilize the structure at low temperature.

119   Similar phenomenon occurs in hematite-luogufengite system (Lee and Xu, 2016). It is also

120   reported that protoenstatite nanocrystals with a large surface area of 615 m^2/g, synthesized using

121   sol-gel and freeze-dry methods, can be quenched to room temperature (Jones et al, 1999).

122   Protoenstatite nanocrystals were synthesized by sol-gel method at 800 °C (Jones et al, 1999),

123   which is lower than the reported phase transition temperature (~1000 °C).

124

Implications

126   The labradorite (An$_{65}$) phenocrysts are very homogeneous in composition (Stewart et al.,

127   1966; Wenk et al., 1980). We infer that the cores of "watermelon" crystals formed at early stages

128   of magma chamber formation at high P-T conditions. The clear phenocryst rims without any

129   precipitates suggest that they formed at a late stage under different conditions. Crystallization of

130   protoenstatite and associated native copper might happen also at a late stage but before magma

131   eruption. The collective effect of the oriented crystals of protoenstatite and clinoenstatite results

132   in the vibrant green colour of "watermelon" sunstones. These results may help understand and

133   determine size-dependent stability of these minerals. We agree that the phenocrysts experienced

134   a thermal shock due to the rapid rising and quenching of the crystals (Hofmeister and Rossman

135   1985). This thermal shock origin can explain why the labradorite crystals without schiller and

136   colors are all cracked. The colored sunstones are thus thermal shock resistant. Like a "single

137   crystal concrete," their nano-inclusions of protoenstatite and Cu probably serve as cushion to

138   absorb the thermal shock due to metallic / plastic behavior of the Cu nanocrystals. This observed

139   texture may inspire the design of new crystalline materials that have strengths to resist thermal

140   shock while being optically clear and colorful.

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142   **ACKNOWLEDGMENTS**

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152   upper-mantle pressures. Nature, 358, 322-324.

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200   **Figures and Captions**

201
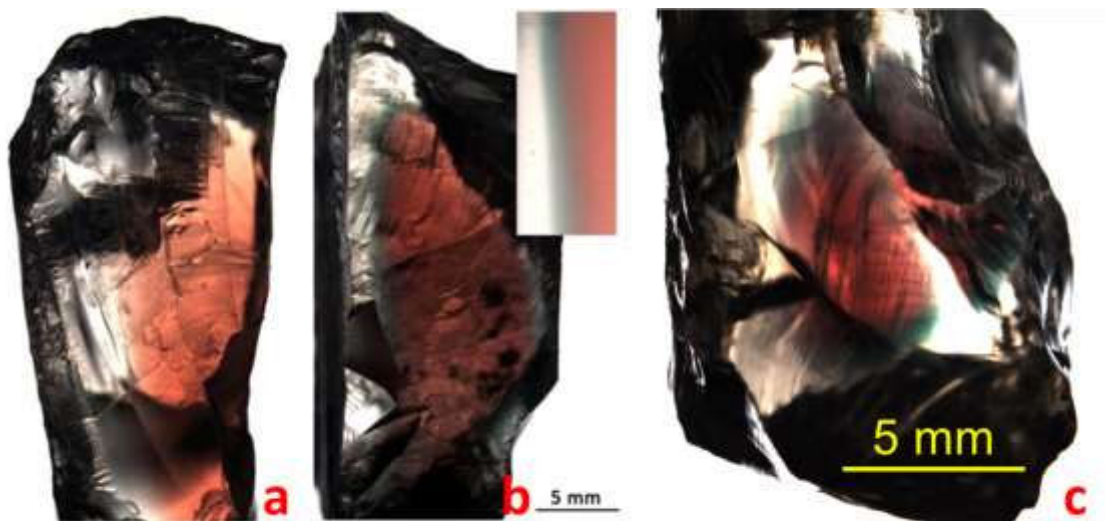
202
203   Figure 1. A gem-quality "watermelon" Oregon sunstone looked down along the normal of (001)

204   (a), and along **b**-axis (b). Note the exceptionally clear rim, along with the zoning of colors. The

205   green border becomes brownish red when looked down along the normal of (001) (a). Top up-

206   right inset shows the detail of the red to green transition from a polished sample (~ 4 mm thick)

207   (b).   Dark areas are due to uneven surfaces that bend the transmitted light away. Another

208   "watermelon" sunstone (c) with a clear rim looked down along ~**b**-axis under a transmitted light

209   from bottom (left). Linear features with dark color in red core and green board are microplates of

210   native Cu.
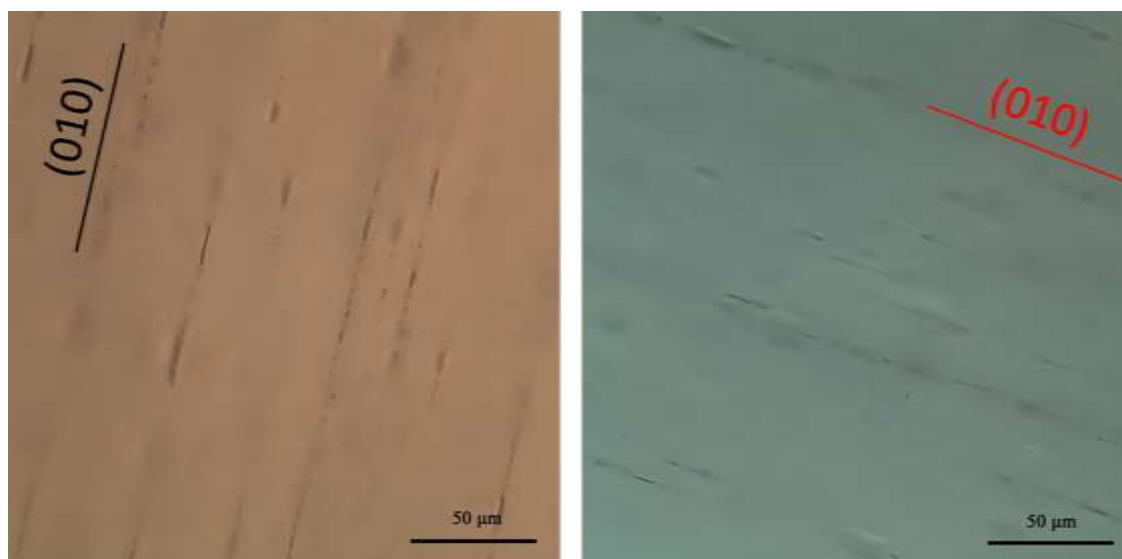
211

212   Figure 2: Transmitted light photomicrographs of a polished sunstone crystal (~ 5 mm thickness)

213   from the green border part show pleochroism (light brown to pale green). The cleavage plane

214   (001) is perpendicular to the light. Some inclusions appear as linear features distributed along the

215   (010) plane of labradorite. The appearance of the inclusions is much larger than their actual sizes

216   due to the strain existing between thenanocrystals and the host labradorite.

217
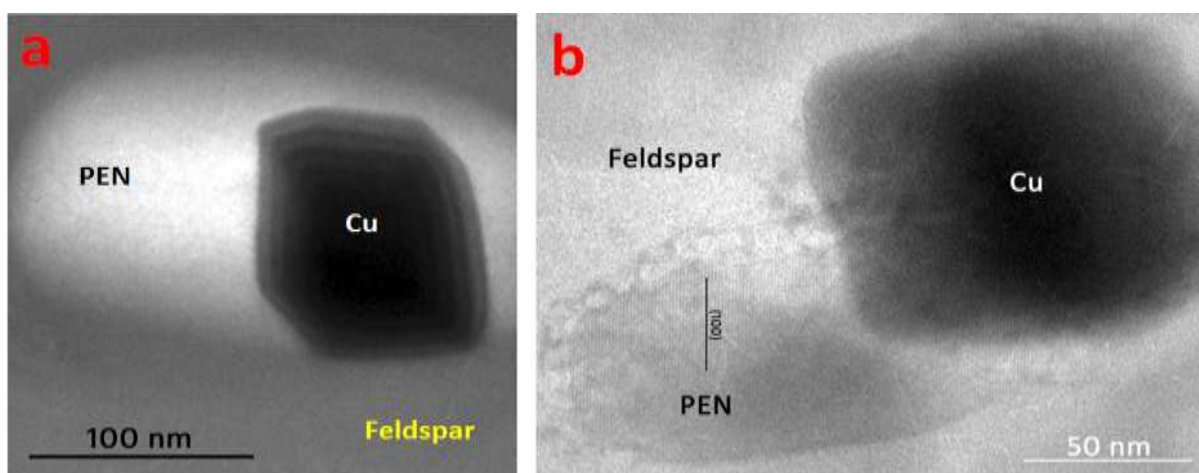
218

219   Figure 3. Dark-field TEM image (a) and bright-field TEM image (b) showing two protoenstatite
220   (PEN) nanocrystals together with native copper nanocrystals (Cu) within labradorite.
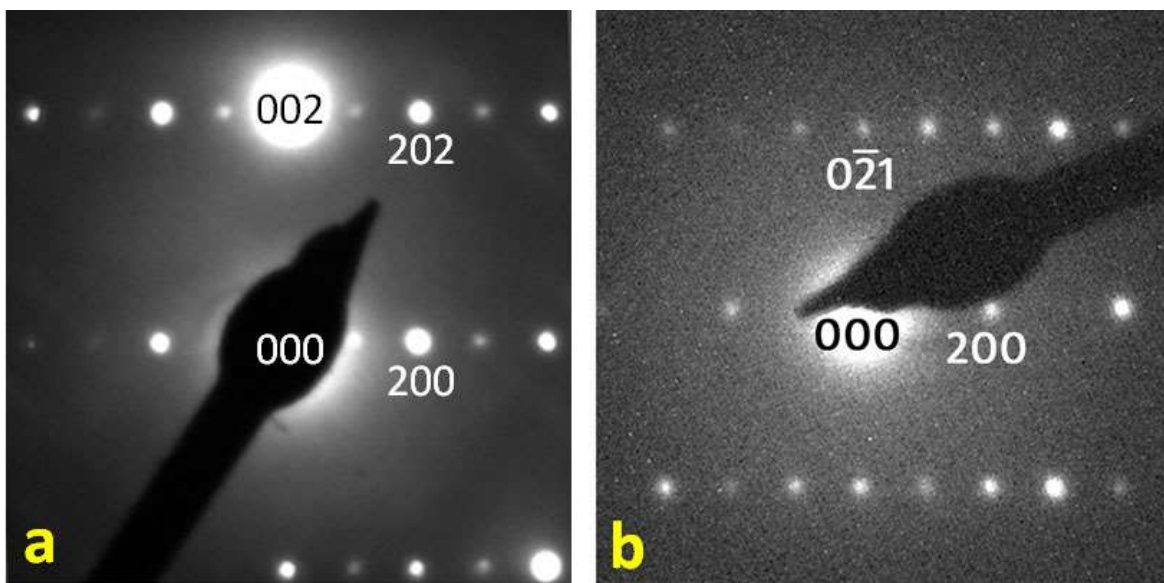
221

222   Figure 4. SAED patterns of protoenstatite nanocrystals shown in Figure 1 along ~ [010] zone-
223   axis (a) and along [012] zone-axis (b). The very strong 002 diffraction spot in the SAED pattern
224   (a) is due to very small excitation error (or diffraction error) for 200, which means that the
225   crystal is very close to two-beam condition. Weak $h00$ ($h = 2n+1$) reflections that violate the 2-
226   fold screw axis symmetry result from multiple diffraction that is common in electron diffraction,
227   especially from thick specimens. The $h00$ ($h = 2n+1$) reflections are extinct from a smaller and
228   thinner crystal (b). The multiple diffraction effect is not obvious here. The SAED patterns
229   confirm the *Pbcn* symmetry.

230
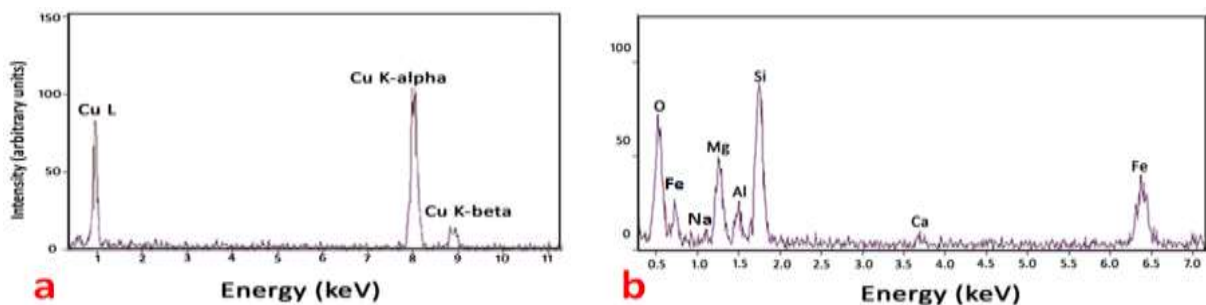
231

232   Figure 5. X-ray EDS spectra from a copper nanocrystal (a) and a protoenstatite nano-crystal (b).
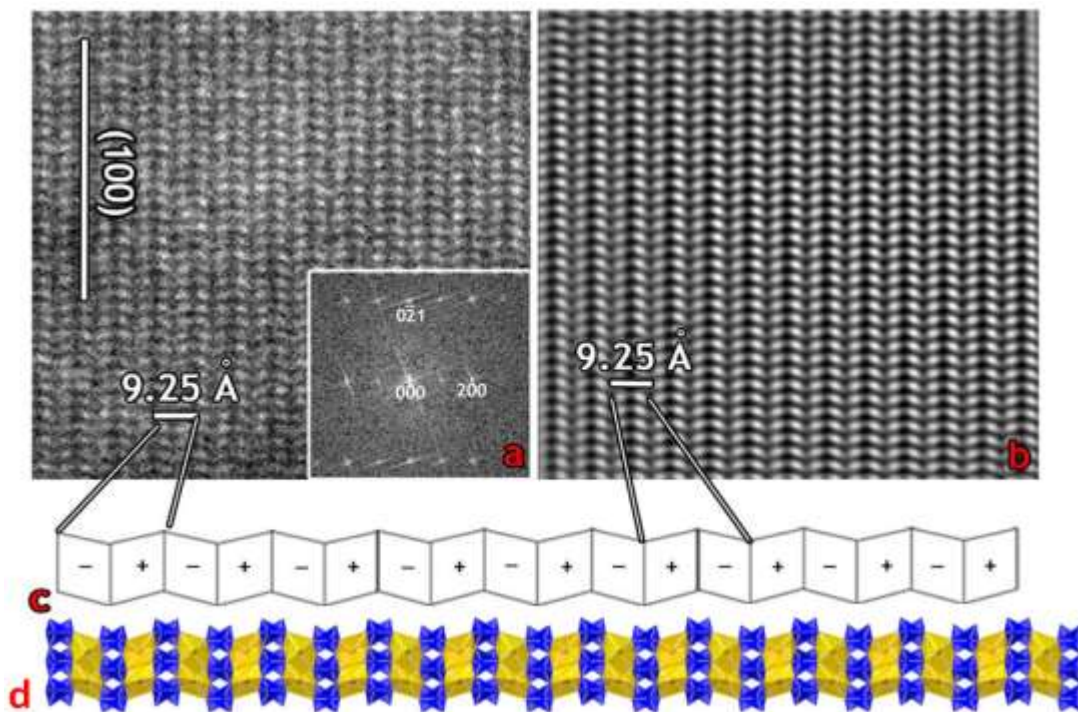
233

234   Figure 6. HRTEM image (a) and noise-filtered HRTEM image (b) of a protoenstatite nanocrystal

235   showing (100) lattice fringes with a periodicity of 9.25 Å. A [012] zone-axis fast Fourier

236   transform (FFT) pattern is inserted at the lower-right corner of the HRTEM image. (c) A simple

237   protoenstatite model based on unit-cell twining of clinoenstatite. (d) A polyhedral model of

238   protoenstatite projected onto (010) showing periodic changes of skews of octahedral layers along

239   the *a*-axis. The model is based on a Li-Sc-bearing protoenstatite at room temperature (Yang et

240   al., 1999) with unit cell parameters and composition measured from the protoenstatite

241   nanocrystals.

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Table.1 Chemical compositions of protoenstatite.

	(1)	(2)	(3)	Average
SiO_2	49.73	51.41	51.60	50.91
TiO_2	0.74	0.37	0.37	0.50
Al_2O_3	11.23	10.23	9.14	10.20
FeO	13.71	15.05	14.05	14.27
MgO	22.57	20.82	22.51	21.97
CaO	0.78	0.73	1.02	0.84
Na_2O	1.24	1.41	1.44	1.37
Si^T	1.78	1.85	1.85	1.82
Ti	0.02	0.01	0.01	0.01
Al^{M1}	0.25	0.28	0.23	0.25
Al^T	0.22	0.15	0.16	0.18
Fe	0.41	0.45	0.42	0.43
Mg	1.20	1.11	1.20	1.17
Ca	0.03	0.03	0.04	0.03
Na	0.09	0.10	0.10	0.09

Table 2: Comparison among all enstatite polymorphs

Phase	Space group	Chain rotation	Skew of octahedra	β angle (°)
Enstatite (En)*	*Pbca*	OA, OB	+ + - - + + - -	90
Clinoenstatite (CEN)*	*P2₁/c*	SA, OB	+ + + +	~ 108
High-CEN**	*C2/c*	O	+ + + +	~ 109
High-P CEN***	*C2/c*	O	+ + + +	~ 101
Protoenstatite (PEN)*	*Pbcn*	O	+ - + -	90
High-P PEN****	*P2₁cn*	SA, OB	+ - + -	90

Notes: * Cameron and Papike 1981; ** Tribaudino et al. (2002); *** Angel et al. (1992); ****
Yang et al. (1999).